

The Crystal Structure of $\text{CaO} \cdot 2\text{Al}_2\text{O}_3$

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The possible use of 0.1% Nd^{3+} in $\text{CaO} \cdot 2\text{Al}_2\text{O}_3$ as a solid-state laser has made it necessary to determine the structure of this compound accurately and eliminate the uncertainty in previous publications. A full X-ray analysis has been performed on a single crystal using a four-circle diffractometer and the results have shown the crystal to have a space group $C2/c$. A comparison with the published structure of the supposed isomorphous strontium compound gave an initial structure which was found to refine satisfactorily to give a reliability index, $R=0.064$, from which the atomic parameters were obtained. The local symmetries of the calcium and aluminum sites are then discussed with respect to the effect on the crystalline electric fields at substituted Nd^{3+} ions.

There has been a certain amount of controversy over the exact crystal structure of the $\text{CaO} \cdot 2\text{Al}_2\text{O}_3$ phase of the calcium aluminate system $\text{CaO}-\text{Al}_2\text{O}_3$. Cockayne (1966) and Cockayne & Robertson (1964) report that this phase is complex hexagonal, whereas Boyko & Wisnyi (1958) report that both $\text{CaO} \cdot 2\text{Al}_2\text{O}_3$ and $\text{SrO} \cdot 2\text{Al}_2\text{O}_3$ form isomorphous monoclinic crystals with a space group $C2/c$ (C_{2h}^6).

In view of the fact that $\text{CaO} \cdot 2\text{Al}_2\text{O}_3$ appears to be a suitable host lattice capable of being doped with small quantities of the Nd^{3+} ion, which is of interest because it forms the basis of a number of the most efficient solid state laser systems, it was decided to attempt an accurate determination of its structure. Once determined one can then make useful estimates of the local symmetry of the substituted Nd^{3+} ions which is essential in calculating the crystalline electric fields which determine the energy levels important to the laser properties. The most important result of our investigation is that the basic structure proposed by Boyko & Wisnyi (henceforth referred to as B.W.) is correct for this phase of the system, while the work of Cockayne appears to be in error.

A number of small crystals, varying in size from about 0.3 mm up to 1 mm, were cut from a single 10 mm dimension crystal grown by the Czochralski technique (Cockayne, 1966). Using a selection of the larger cut crystals, the space group was determined by taking a number of X-ray photographs on a single-crystal precession camera using $\text{Cu } K\alpha$ radiation. From the exposed plates it was a simple matter to identify the three principal zones $hk0$, $h0l$ and $0kl$. Of the possible unit cell types proposed by previous authors only a monoclinic unit cell with $\beta=107^\circ$ fitted the observed symmetry. In addition a photograph of the entire $h0l$ zone was taken on a Weissenberg camera, which clearly showed that we had chosen the most sensible monoclinic axes having $\beta=107^\circ$, and that it was impossible to fit the observed symmetry to a hexagonal type structure involving an angle of 120° , as proposed by Cockayne.

The systematic absences observed on the plates indicated that the space group could be either Cc (C_2^4) or $C2/c$ (C_{2h}^6). Since the plates showed definite evidence of absorption of the $\text{Cu } K\alpha$ radiation, additional data were collected on a Hilger & Watts computer controlled four circle X-ray diffractometer using $\text{Mo } K\alpha$ radiation. The linear absorption coefficient for $\text{CaO} \cdot 2\text{Al}_2\text{O}_3$ using $\text{Cu } K\alpha$ radiation is $\mu=143 \text{ cm}^{-1}$, compared with $\mu=15.8 \text{ cm}^{-1}$ for $\text{Mo } K\alpha$ radiation. Hence by using $\text{Mo } K\alpha$ radiation the effects of absorption could be very much reduced. However, it was not possible to make an accurate assessment of the correction factor for the measured intensities because of the irregular shape of the crystal. This does not appear to be serious as an estimate of the correction for a sphere of equivalent size (0.3 mm across) indicated variations in intensity of some 2% compared with experimental errors of about 5%.

The positions of 14 strong low-angle X-ray reflexions were measured manually on the diffractometer and were then used as the basis to determine a set of accurate unit-cell parameters by a least-squares fit starting with the parameters of the monoclinic unit cell from precession photographs. The results of this fit are shown in Table 1 together with the previously published values of B.W. The errors quoted are from the least-squares fit which are rather optimistic, and the actual standard deviations are probably about 3 or 5 times greater.

Table 1. Unit cell parameters of $\text{CaO} \cdot 2\text{Al}_2\text{O}_3$

	This work	B.W.
a	$12.8398 \pm 0.0057 \text{ \AA}$	12.89 \AA
b	8.8624 ± 0.0032	8.88
c	5.4311 ± 0.0018	5.45
α	90°	90°
β	$106^\circ 50' \pm 2'$	$107^\circ 3'$
γ	90°	90°

Using the data of Table 1 to define the unit cell of the crystal, the computer control of the diffractometer was set to perform a step scan of ω and θ over all pos-

sible X-ray reflexions within a diffraction angle 2θ from 0 to 80° and within the two octants of diffraction indices defined by h, k, l and \bar{l} . A total of 1901 reflexions were measured in this way of which only about ten pairs were crystallographically equivalent.

A set of standard computer programs for the University of York, Elliott 4130 computer, was then employed to analyse the experimental data to produce structure factors from the integrated X-ray intensities.

The close similarity between our results and those of B.W. for the unit cell data, as shown on Table 1 makes it seem likely that their interpretation of the structure would form a suitable basis for interpreting our structure factors. B.W. published a set of atomic parameters for the structure of $\text{SrO} \cdot 2\text{Al}_2\text{O}_3$ which they concluded was isomorphous with the calcium compound. By using these values as a starting point we performed a full-matrix fit to the 1901 measured structure factors on the University of York computer with a modified version of the Oak Ridge Fortran least-squares program of Busing, Martin & Levy (1962).

Approximately 55 structure factors were unobserved on analysing the intensity data, but in view of the large number correctly observed these few were omitted in the least-squares fit. A Rollett weighting scheme was used to weight each reflexion in the fitting program. The appropriate ionic scattering factors for the calculation were taken from *International Tables for X-ray Crystallography* (1965), where the ionization states were assumed to be Ca^{2+} , Al^{3+} and O^{2-} .

With isotropic temperature factors for all the atoms the proposed structure was found to refine satisfactorily, and produced a reliability index of $R(hkl) = 0.093$. On conversion to anisotropic temperature factors the structure was refined even further to produce a minimum value of $R = 0.064$ at which point the calculated variations in the parameters were smaller than their standard deviations. The satisfactory low value of $R(hkl)$ indicates that the structure is correct and that the space group is $C2/c(C_{2h}^6)$ as proposed by B.W. The refined atomic parameters are given in Table 2. Both the positions x, y, z and the anisotropic temperature factors, β , are included. For completeness the

general and special positions for the space group are listed at the bottom of the Table.

There are four formula units $\text{CaO} \cdot 2\text{Al}_2\text{O}_3$ in a unit cell, and the calcium atom and one of the oxygen atoms, O(1), lie on special positions. The volume of the unit cell is 591.54 \AA^3 and so the calculated density is 2.915 g.cm^{-3} .

The observed structure factors (FO), and the calculated structure factors (FC) based on the parameters of Table 2 are listed in Table 3.

In order to assess the likely positions of dopant Nd^{3+} ions which were in fact present to the amount of 0.1% by weight in the measured sample, it is only necessary to consider the calcium and aluminum ions, as they have similar electric charge to Nd^{3+} . The calcium ions have a point group 2, with the axis along the crystallographic b axis. Each has seven nearest neighbour oxygen ions varying in distance between 2.33 and 2.88 Å , which are only about 5° away from forming a point group $2mm (= C_{2v})$.

The aluminum sites, however, have no well defined symmetry and the local point group is 1 ($= C_1$). Despite this the two inequivalent aluminum sites are at the centre of tetrahedra of oxygen ions which are very close to regular tetrahedra. The aluminum-oxygen bond distances and bond angles for the two sites are given in Table 4.

For a perfect tetrahedron the bond angles would all be equal to 109.46° , which shows that these sites are slightly distorted but will experience a predominantly cubic crystal field. Nevertheless, a point charge calculation for both these sites indicates that all crystal field terms up to V_6^0 will be required to describe the splitting of the magnetic energy levels, if the Nd^{3+} ion enters the lattice at these points.

An unambiguous assignment of the likely positions of the dopant Nd^{3+} ions is not possible from the structure alone. If the Nd^{3+} enters the calcium site the difference in charge, as calcium forms Ca^{2+} ions, will necessitate some form of charge compensation in the lattice similar to that required in the laser material Nd^{3+} in CaWO_4 . If the Nd^{3+} enters at the aluminum sites no charge compensation would be necessary as aluminum forms Al^{3+} ions. A final assignment can

Table 2. *The atomic positional and thermal parameters of $\text{CaO} \cdot 2\text{Al}_2\text{O}_3$*

	Ca	O(1)	Al(1)	Al(2)	O(2)	O(3)	O(4)
X	0	0	0.1641	0.1198	0.1155	0.1185	0.1924
Y	0.8091	0.5231	0.0867	0.4406	0.0512	0.2553	0.4436
Z	0.25	0.25	0.3030	0.2410	0.5659	0.1491	0.5797
β_{11}	0.00147	0.00142	0.00122	0.00124	0.00169	0.00200	0.00130
β_{22}	0.00252	0.00228	0.00205	0.00200	0.00310	0.00210	0.00325
β_{33}	0.00809	0.01510	0.00715	0.00876	0.00990	0.00938	0.00839
β_{12}	0	0	0.00007	0.00001	0.00043	0.00016	0.00020
β_{13}	0.00173	0.00297	0.00183	0.00220	0.00261	0.00158	0.00177
β_{23}	0	0	0.00007	0.00012	0.00130	0.00022	0.00037
Positional error:	± 0.0010						
Temperature factor error:	± 0.0005						
General position:	$x, y, z; \bar{x}, \bar{y}, \bar{z}; \bar{x}, y, \frac{1}{2} - z; x, \bar{y}, \frac{1}{2} + z; \frac{1}{2} + x, \frac{1}{2} + y, z; \frac{1}{2} - x, \frac{1}{2} - y, \bar{z}; \frac{1}{2} - x, \frac{1}{2} + y, \frac{1}{2} - z; \frac{1}{2} + x, \frac{1}{2} - y, \frac{1}{2} + z$						
Special position:	$0, y, \frac{1}{4}; 0, \bar{y}, \frac{3}{4}; \frac{1}{2}, \frac{1}{2} + y, \frac{1}{4}; \frac{1}{2}, \frac{1}{2} - y, \frac{3}{4}$						

Table 3 (cont.)

H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC			
6	92	-92	5	449	506	18	9	15	8	76	-73	9	111	122	13	J	2	12	35	-32												
8	102	-102	7	83	-93							11	138	-144																		
10	89	-86	9	160	-168	K = 3 L = 3	K = 13 L = 3	K = 13 L = 3	13	61	-62	K = 6 L = 5	K = 5 L = 6	K = 5 L = 6																		
K = 15 L = 1	11	210	214	1	270	276	1	175	-177	15	76	-22	0	224	222	1	299	-293	0	94	92											
1	55	-55	13	156	163	3	474	-530	3	81	77	2	136	140	3	25	-29	4	65	-59												
3	18	-6	17	31	-22	5	131	145	5	116	115	K = 8 L = 4	4	99	-99	5	13	9	6	110	97											
5	182	-176	7	84	98	7	84	98	7	95	-95	0	353	-355	6	34	41	7	162	-168												
K = 0 L = 2	9	100	116	11	139	150	K = 14 L = 3	K = 14 L = 3	6	89	-86	10	71	73	11	22	-20	1	109	102												
0	879	H	0	329	355	13	186	-197	0	219	220	10	118	-122	K = 7 L = 5	K = 6 L = 6	K = 6 L = 6	0	60	-56	K = 10 L = 7											
2	653	-786	4	50	-52	15	21	22	2	55	-48	12	51	49	1	83	-77	0	60	-56												
4	14	5	6	206	216	17	204	200	4	64	62	14	58	-54	3	153	151	2	171	-174	0	150	138									
6	290	-329	8	195	206	K = 4 L = 3	K = 4 L = 3	K = 4 L = 3	6	152	146	K = 9 L = 4	K = 9 L = 4	K = 9 L = 4	7	13	1	6	82	-84												
8	172	-191	10	76	82	0	95	90	1	87	78	3	22	-18	9	50	49	8	106	-105	K = 0 L = 8											
10	277	-323	12	113	114	2	100	98	K = 15 L = 3	K = 15 L = 3	K = 15 L = 3	5	150	-149	11	128	124	10	47	43	0	54	45									
12	297	-335	14	20	20	6	242	240	1	79	-78	5	43	-43	13	67	62	K = 8 L = 5	1	110	-115	K = 2 L = 8										
14	132	144	16	116	114	8	46	-44	K = 0 L = 4	K = 0 L = 4	K = 0 L = 4	9	125	130	K = 8 L = 5	1	110	-115	4	90	85											
16	124	-135	K = 9 L = 2	12	297	313	0	476	472	0	27	-38	11	26	-25	K = 8 L = 5	1	110	-115	4	90	85										
18	289	-298	1	161	-166	14	59	63	2	27	-38	11	26	-25	K = 8 L = 5	1	110	-115	4	90	85											
20	19	16	3	217	233	16	19	13	4	306	338	13	81	-77	4	16	-5	5	114	113	K = 1 L = 8											
K = 1 L = 2	5	186	188	18	148	138	6	126	143	8	127	145	K = 10 L = 4	0	73	67	10	43	42	1	190	-180										
1	488	534	7	168	-176	K = 5 L = 3	10	246	276	10	246	276	K = 10 L = 4	0	73	67	10	43	42	0	318	319										
3	97	-108	9	58	-55	K = 5 L = 3	10	246	276	10	246	276	K = 10 L = 4	0	73	67	10	43	42	0	318	319										
5	607	-727	11	62	69	1	398	400	12	61	65	4	140	151	12	38	-37	K = 8 L = 6	0	318	319											
7	309	348	13	101	98	3	202	-205	14	17	21	6	144	148	12	38	-37	K = 8 L = 6	0	318	319											
9	324	353	15	27	15	5	109	-119	16	138	140	8	67	-64	K = 9 L = 5	2	141	141	K = 2 L = 8													
11	232	-269	K = 10 L = 2	9	157	172	K = 1 L = 4	1	355	-363	K = 1 L = 4	1	355	-363	12	133	120	3	124	-124	8	113	109									
13	165	-176	0	104	-106	11	138	145	1	355	-363	12	133	120	3	124	-124	8	113	109	2	49	49									
15	82	87	2	209	209	13	182	-193	3	103	-106	K = 11 L = 4	1	60	67	K = 9 L = 5	2	141	141	K = 2 L = 8												
17	83	82	4	240	-247	15	34	-32	5	407	474	1	60	67	K = 9 L = 5	2	141	141	K = 2 L = 8													
19	58	-54	6	270	-275	17	126	119	7	40	-43	3	109	113	K = 10 L = 5	3	82	76	K = 3 L = 8													
K = 2 L = 2	8	59	65	K = 6 L = 3	11	144	158	13	41	41	9	116	111	4	86	82	K = 10 L = 5	3	82	76	K = 3 L = 8											
0	20	-11	10	16	-14	0	332	-318	13	5	-4	9	116	111	4	86	82	K = 10 L = 5	3	82	76	K = 3 L = 8										
2	160	185	12	111	-107	0	332	-318	13	5	-4	9	116	111	4	86	82	K = 10 L = 5	3	82	76	K = 3 L = 8										
4	296	340	14	107	-100	2	20	-26	15	5	-4	9	116	111	4	86	82	K = 10 L = 5	3	82	76	K = 3 L = 8										
6	76	-89	4	19	-12	4	19	-12	17	47	43	K = 12 L = 4	0	73	-74	K = 10 L = 5	3	82	76	K = 3 L = 8												
8	69	69	K = 11 L = 2	6	108	-117	K = 2 L = 4	0	205	202	4	80	78	K = 11 L = 5	4	80	78	K = 11 L = 5	4	80	78	K = 11 L = 5										
10	121	143	1	226	-223	8	70	-70	K = 2 L = 4	0	205	202	4	80	78	K = 11 L = 5	4	80	78	K = 11 L = 5												
12	19	12	3	82	84	10	132	-139	0	326	-338	6	20	-18	3	83	79	K = 11 L = 5	4	80	78	K = 11 L = 5										
14	134	144	5	100	-100	12	10	-5	2	326	-338	6	20	-18	3	83	79	K = 11 L = 5	4	80	78	K = 11 L = 5										
16	8	15	7	182	-190	14	16	-9	4	208	-212	8	186	-174	5	40	35	K = 11 L = 5	4	80	78	K = 11 L = 5										
18	18	-21	9	66	-66	16	117	-110	6	73	-82	8	68	78	K = 13 L = 4	1	121	-116	K = 12 L = 5	0	9	4										
20	127	115	11	133	-132	10	15	-15	12	171	-194	3	80	-74	K = 14 L = 4	0	71	14	K = 12 L = 5	0	9	4										
K = 3 L = 2	1	497	535	K = 12 L = 2	3	276	-297	14	70	-77	16	45	44	K = 14 L = 4	0	71	14	K = 12 L = 5	0	9	4											
1	497	535	0	83	-82	5	326	-366	16	45	44	K = 14 L = 4	0	71	14	K = 12 L = 5	0	9	4													
3	99	105	2	157	166	7	35	-30	K = 3 L = 4	0	71	14	K = 12 L = 5	0	9	4																
5	93	97	4	29	29	9	32	31	K = 3 L = 4	0	71	14	K = 12 L = 5	0	9	4																
7	370	399	6	16	-15	11	199	-214	3	308	-297	2	77	-72	K = 13 L = 5	3	98	-99	K = 1 L = 7	1	220	-209										
9	219	251	8	123	124	13	179	-185	5	63	-68	K = 1 L = 5	1	254	261	K = 0 L = 6	0	340	-338	K = 1 L = 7	1	220	-209									
11	106	112	10	10	-3	K = 8 L = 3	7	108	-120	11	203	-231	5	89	97	0	340	-338	K = 1 L = 7	1	220	-209										
13	62	65	K = 13 L = 2	2	285	309	11	103	-109	5	54	60	2	73	-71	K = 0 L = 6	0	340	-338	K = 1 L = 7	1	220	-209									
15	102	110	1	111	115	6	88	-87	13	54	-58	7	177	192	4	166	-171	K = 0 L = 6	0	340	-338	K = 1 L = 7	1	220	-209							
17	179	178	5	9	-9	8	177	186	15	69	71	9	127	136	6	58	-62	K = 0 L = 6	0	340	-338	K = 1 L = 7	1	220	-209							
19	65	59	7	46	41	12	10	-13	17	87	-82	11	22	23	8	59	-65	K = 0 L = 6	0	340	-338	K = 1 L = 7	1	220	-209							
K = 4 L = 2	4	462	-485	9	131	127	14	93	93	K = 4 L = 4	0	65	-61	13	82	88	10	250	-265	K = 0 L = 6	0	340	-338	K = 1 L = 7	1	220	-209					
2	464	-499	K = 14 L = 2	0	14	5	K = 9 L = 3	4	25	21	0	272	275	1	239	243	K = 0 L = 6	0	340	-338	K = 1 L = 7	1	220	-209								
4	199	210	0	14	5	K = 9 L = 3	4	25	21	0	272	275	1	239	243	K = 0 L = 6	0	340	-338	K = 1 L = 7	1	220	-209									
6	156	-162	1	196	1																											

Table 3 (cont.)

H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC	H	FO	FC	
7	441	-385	K = 14	L = -3	6	136	128	1	110	94	14	64	63	10	20	-15	12	344	345	K = 5	L = 1	1	129	-142
9	190	-178	0	214	-220	8	207	194	3	948	-983	16	111	107	12	131	-136	14	155	150	3	276	293	
11	234	222	2	117	-119	10	425	-416	5	664	-625	18	54	-52	14	186	-183	16	133	-125	5	215	233	
13	201	191	4	56	54	12	183	-172	7	164	151	16	15	13	18	14	16	K = 11	L = 0	7	176	-183		
15	193	-178	6	123	-123	14	31	31	9	288	-268	K = 9	L = -1	18	73	-72	K = 11	L = 0	5	217	-238			
17	121	-117	8	138	-141	16	129	-123	11	322	-291	1	71	-73	20	73	-72	1	269	281	7	176	-183	
19	46	50	10	44	-46	20	.94	-92	13	166	-162	3	281	292	22	109	-98	5	126	-123	9	217	-238	
21	79	86	K = 15	L = -3	K = 7	L = -2	17	30	-22	7	38	-37	K = 3	L = 0	7	213	214	13	110	110	11	117	126	
K = 6	L = -3	1	117	-113	1	197	189	19	76	-74	9	253	250	1	499	-521	9	39	15	15	31	-22		
0	325	318	3	129	-129	3	54	48	21	192	-181	11	77	70	3	260	-263	11	127	128	17	120	-115	
2	37	18	K = 0	L = -2	5	338	317	K = 2	L = -1	13	40	-37	5	45	-46	13	92	89	19	49	-39			
4	29	-23	0	864	H	9	308	-302	0	117	-115	17	73	67	9	250	-251	K = 12	L = 0	0	435	446		
6	202	177	2	9	9	11	175	166	2	309	-283	K = 10	L = -1	13	103	-102	0	42	50	2	233	236		
8	149	137	0	864	H	9	308	-302	0	117	-115	17	73	67	9	250	-251	2	115	-107	4	84	-88	
10	127	115	4	167	150	13	357	341	4	115	106	K = 10	L = -1	13	103	-102	0	42	50	2	233	236		
12	26	-30	6	897	-849	15	49	-47	6	30	37	0	60	55	15	185	-182	2	115	-107	4	84	-88	
14	79	72	8	997	H	17	73	-67	8	546	-517	2	170	178	17	145	-141	4	121	-118	6	126	-139	
16	157	152	10	225	203	19	98	99	10	315	-286	4	116	114	19	83	-76	8	80	-83	8	278	303	
18	9	13	12	35	-22	12	259	254	6	17	-25	21	89	-82	10	144	-139	10	200	211	12	51	-50	
20	30	32	14	449	-433	K = 8	L = -2	16	368	-349	8	191	189	K = 4	L = 0	0	287	279	14	38	-43	14	8	-6
K = -7	L = -3	1	136	141	20	45	-41	4	9	-6	22	107	-97	14	40	39	2	224	228	K = 13	L = 0	1	81	-84
3	275	281	22	36	36	6	114	113	8	380	377	K = 3	L = -1	6	173	-173	3	132	-134	K = 7	L = 1	1	436	454
5	272	272	K = 1	L = -2	10	107	103	1	796	879	K = 11	L = -1	8	286	292	5	18	16	1	436	454			
7	314	306	1	79	-72	12	10	12	3	116	-111	1	283	-292	10	230	228	7	54	-56	3	135	129	
9	80	-76	3	327	288	14	157	155	5	258	-241	3	75	75	12	180	-185	9	186	-186	5	289	283	
11	95	86	5	555	-499	16	103	99	7	432	410	5	51	54	14	5	8	11	33	-29	7	122	132	
13	473	447	7	280	-247	18	156	150	9	168	-163	7	275	-277	16	158	149	13	18	20	9	17	14	
15	79	72	9	747	708	20	44	50	11	142	132	9	145	-151	18	8	1	11	328	340				
17	35	-41	11	131	-113	13	172	169	11	30	-19	K = 5	L = 0	0	30	-35	13	149	142	13	149	142		
19	155	156	13	453	-434	K = 9	L = -2	15	223	-210	13	20	-14	K = 5	L = 0	0	30	-35	15	35	-35			
K = 8	L = -3	15	127	121	1	163	-168	17	144	132	15	61	-57	1	104	105	2	86	-90	17	140	132		
0	66	66	17	180	170	3	96	93	19	153	144	3	460	473	4	77	-79	4	77	-79	19	133	121	
2	39	-42	19	53	-55	5	280	288	21	95	-89	K = 12	L = -1	0	94	98	7	10	18	8	78	-74		
4	273	-269	21	107	-102	7	63	-56	0	94	98	2	102	-97	9	440	430	10	71	-71	0	187	173	
6	24	13	23	18	11	9	233	-224	K = 4	L = -1	2	102	-97	9	440	430	10	71	-71	0	187	173		
8	214	-201	11	116	109	0	346	354	4	343	-349	11	56	56	11	56	56	2	228	-228				
12	1	-3	K = 2	L = -2	13	122	119	2	308	300	8	87	90	13	48	-52	K = 15	L = 0	4	130	-135			
14	35	30	0	19	-11	15	27	25	4	766	790	10	128	-129	15	252	243	1	22	21	6	62	67	
16	154	-151	2	57	64	17	65	-63	6	476	455	12	147	-145	17	83	80	3	145	153	8	146	-158	
18	30	-30	4	297	279	19	68	-64	8	188	-181	14	109	-106	19	21	-2	5	86	79	10	79	-81	
20	34	34	6	217	183	10	126	115	10	126	115	10	126	115	21	129	116	7	41	47	12	74	73	
K = 9	L = -3	10	122	206	0	101	-106	14	298	277	1	75	-77	K = 6	L = 0	0	640	-697	0	142	144	16	96	-92
1	176	180	12	184	168	2	264	-269	16	166	-157	3	112	-116	0	640	-697	0	142	144	16	96	-92	
3	183	-182	14	21	7	4	97	-89	18	113	110	5	94	96	2	366	368	K = 1	L = 1	K = 9	L = 1	1	101	-106
5	286	-283	16	119	109	6	142	-138	20	296	279	9	117	-118	4	679	729	1	86	93	3	310	-329	
7	53	50	18	32	-27	8	86	79	22	52	46	11	60	-48	6	103	-106	1	86	93	3	310	-329	
9	111	-108	20	69	63	10	76	-69	K = 5	L = -1	1	558	585	K = 14	L = -1	14	237	226	7	139	147	7	30	-63
11	153	-146	22	140	134	12	307	-308	5	377	-382	0	254	266	16	15	18	9	119	120	9	267	-264	
13	11	9	K = 3	L = -2	1	537	543	3	422	-422	2	84	90	18	84	-80	11	135	150	11	15	-14		
15	141	-138	3	395	363	K = 11	L = -2	7	410	404	4	123	-127	20	136	125	13	197	210	15	147	-141		
17	107	-104	5	74	60	1	322	-335	9	88	75	6	198	202	19	38	39	15	181	187	17	35	-28	
19	42	43	7	365	327	3	46	35	11	68	-64	8	180	178	K = 7	L = 0	1	54	-53	21	175	164		
K = 10	L = -3	9	519	479	5	39	42	13	70	-75	10	23	-17	1	54	-53	K = 10	L = 0	2	53	-55			
0	142	-149	9	519	479	5	39	42	13	70	-75	10	23	-17	1	54	-53	K = 10	L = 0	2	53	-55		
2	33	-31	11	164	148	7	296	-293	15	72	-67	K = 15	L = -1	5	275	-269	K = 2	L = 1	2	207	-214			
4	144	-139	13	108	99	9	148	-156	17	179	176	1	49	49	7	148	-157	0	117	115	4	69	-76	
6	38	-36	15	152	146	11	132	-4	19	14	9	1	14	22	9	199	206	2	581	660	6	62	57	
8	91	-81	17	205	188	13	139	-133	21	116	-110	3	14	22	11	142	-146	4	74	74	8	249	-250	
10	249	-248	19	173	167	15	37	-34	K = 6	L = -1	5	148	148	13	307	-300	6	424	-468	10	134	-136		
12	19	22	21	49	29	0	434	-446	K = 4	L = -1	7	69	64	15	112	104	8	523	571	12	78	73		
14	30	21	K = 12	L = -2	0	634	-446	2	95	-92	K = 2	L = 0	0	205	210	19	139	-130	10	306	311	14	107	-102
16	211	-205	K = 4	L = -2	0	82	-82	2	95	-92	K = 2	L = 0	0	205	210	19	139	-130	10	306	311	14	107	-102
18	67	-62	0	83	51	2	48	44	4	397	400	6	266	-251	4	729	-780	K = 8	L = 0	14	109	114		
K = 11	L = -3	4	245	-227	6	76	-78	8	438	-421	6	721	763	0	632	-639	16	150	145	K = 11	L = 1	1	218	226
1	182	186	6	102	-84	8	19	22	10	15	2	8	662	693	2	85	-85	18	31	33	1	218	226	
3	103	-102	8	140	-128	10	156	154	12	64	61	10	406	413	4	97	94	20	73	65	5	32	-30	
5	44	52	10	104	-98	12	19	-26	14	9	2	12	51	46	6	237	-248	7	144	144	7	144	144	
7	236	240	12	135	127	14	30	21	16	207	-201	14	74	-82	8	225	-236	K = 3	L = 1	9	191	202		
9	135	139	14	78	-72	18	137	-132	18	321	315	16	321	315	10	215	-							

Table 4. Al-O bond distances and angles

Al-O bond distances			
Al(1)-O(3)	1.729 Å	Al(2)-O(3)	1.716 Å
Al(1)-O(2)	1.744	Al(2)-O(1)	1.752
Al(1)-O(2)†	1.753	Al(2)-O(4)†	1.779
Al(1)-O(4)*	1.789	Al(2)-O(4)	1.804
Bond angles			
O(3)-Al(1)-O(2)	113.64°	O(3)-Al(2)-O(1)	121.07°
O(3)-Al(1)-O(2)†	104.37	O(3)-Al(2)-O(4)†	111.32
O(3)-Al(1)-O(4)*	118.18	O(3)-Al(2)-O(4)	105.64
O(2)-Al(1)-O(2)†	109.89	O(1)-Al(2)-O(4)†	109.63
O(2)-Al(1)-O(4)*	104.83	O(1)-Al(2)-O(4)	99.38
†O(2)-Al(1)-O(4)*	105.55	†O(4)-Al(2)-O(4)	107.75

The superscripts on the oxygen atoms indicate the positions relative to the basis atoms in Table 2.

* Position $\frac{1}{2} - x, \frac{1}{2} - y, \bar{z}$.

† Position $x, \bar{y}, \frac{1}{2} + z$.

only be achieved after additional electron spin resonance and optical measurements have been performed on the doped sample, when the exact symmetry of the sites should become obvious.

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A Study of the Crystal structure of β -Cyclotetramethylene Tetranitramine by Neutron Diffraction

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The crystal structure of β -cyclotetramethylene tetranitramine has been reinvestigated by means of neutron diffraction. Least-square refinement of all positional and anisotropic thermal parameters with 545 non-zero reflections yielded a final R value of 0.059. The heavy-atom parameters obtained agree with those from X-ray determinations except for slight position shifts of a few atoms. The positions of the hydrogen atoms have been determined for the first time. All hydrogen atoms are located close to nearby oxygen atoms, a few of which form intramolecular or intermolecular hydrogen bonds of the type C-H...O. Several short intramolecular and intermolecular distances between oxygen and other atoms have been measured.

Introduction

Cyclotetramethylene tetranitramine, known as HMX, is a well-known explosive and a high-melting-point by-product in the manufacture of RDX. β -HMX is the

room temperature stable phase of the four known polymorphic forms, whose crystallographic data are summarized in the paper of Cady, Larson & Cromer (1963). The positions of the heavy-atoms were reported by Eiland & Pepinsky (1955) from a three-dimensional X-ray investigation using an isotropic bulk temperature factor. The same data were further refined by Cady, Larson & Cromer (1963) using anisotropic temperature factors, but hydrogen positions were not reported. In the present work, the structure is reinvestigated to

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